

DESCRIPTION

HIGH STRENGTH ELECTRICAL STEEL SHEET AND
PROCESSED PART OF SAME AND
5 METHODS OF PRODUCTION OF SAME

TECHNICAL FIELD

The present invention provides a high strength
electrical steel sheet, in particular a non-oriented
10 electrical steel sheet, containing Cu appropriately
treated to form a fine Cu metal phase maintaining good
magnetic properties. The electrical steel sheet obtained
by the present invention is especially suited to its use
in high speed rotary machines requiring strength,
15 electromagnetic switches requiring wear resistance, etc.

BACKGROUND ART

Until recently the rotational speed required in
rotary devices was at most been about 100,000 rpm and
laminated electrical steel sheet was used for the
20 material of rotor cores. Recently, superhigh speed
rotation as high as 200,000 or 300,000 rpm has been
demanded and thus the centrifugal force applied to the
rotor could exceed the strength of the electrical steel
sheet. Further, inner magnet type motors are increasingly
25 used and the load applied to the material of the rotor
during rotation becomes large. Thus, the fatigue strength
of the material is becoming important in many cases.

In other cases, the contact surface of
electromagnetic switch is worn during its use, and thus
30 magnetic materials superior in not only electromagnetic
properties, but also wear resistance are desired for the
use.

To meet with these needs, recently high strength
non-oriented electrical steel sheet has been studied.
35 Several proposals have been made. For example, Japanese
Published Patent Application No. 1-162748 and Japanese
Published Patent Application No. 61-84360 propose a

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material using a slab increased in Si content and further containing one or more of Mn, Ni, Mo, Cr, and other solid solution strengthening components, but the sheet is liable to break easily in rolling and causes less productivity and less yield. Thus the sheet has a room for improvement. Furthermore, since Ni, Mo or Cr are included in large amounts in the steel, the material becomes extremely expensive.

Japanese Published Patent Application No. 61-87848 discloses producing high strength non-oriented electrical steel sheet by rapid solidification from a melt containing 2.5% or more of Si. Japanese Published Patent Application No. 8-41601 discloses improving the rollability by wrapping a high Si steel containing 2.5% or more Si by low Si steel containing 2.0% or less Si. Since these proposals use special processes, the sheets cannot be produced by the production facilities for conventional electrical steel sheet and therefore are difficult to be produced industrially.

With the above methods utilizing solid solute strengthening by solute elements, from the view point of magnetic properties, the saturation magnetic flux density of the material is inherently low, and thus the magnetic flux density of the product sheet is inevitably low. Further, from the view point of crystal structure, the methods inherently refines grain size, so while these are preferable in terms of increasing the strength, there is the problem that the core loss ends up rising.

Further, to strengthen a material, utilizing precipitates may also be considered, but precipitates also end up degrading the magnetic properties from the viewpoint of the magnetic flux density and core loss due to the effects of the precipitates themselves and the refining the crystal structure. In this way, high strength electrical steel sheets have inherent problems wherein the magnetic properties originally required are remarkably degraded.

In particular, with materials strengthened by refining grain size or by precipitates, when punched to an article for electrical appliances such as motors etc., in the stress relief annealing (SRA) process for
5 relieving the fabrication stress introduced to the steel sheets, growth of the crystal structure or precipitates occurring while holding the steel at a high temperature is unavoidable and therefore the strength is decreased. Further, use of high strength materials accelerates the
10 wear of the dies when punching the steel into parts for electrical appliances, in particular in the shearing process, so becomes a cause of raising the cost of production of the electrical appliances.

DISCLOSURE OF THE INVENTION

15 In this way, various proposals have been made regarding high strength electrical steel sheet, but the fact is that it is not yet possible to stably produce such steel industrially securing the required magnetic properties and using an conventional electrical steel
20 sheet production facility. Further, there are also many remaining problems such as softening in the stress relief annealing performed after fabrication or the wear of the dies during punching parts for electrical appliances.

The object of the present invention is to stably
25 produce high strength electrical steel sheet which is high in strength and wear resistance and is superior in magnetic properties such as magnetic flux density and core loss without greatly changing the productivity such as cold rollability from that of conventional electrical
30 steel sheet production process.

Another object is the production of an electrical steel sheet which is relatively soft until the completion of stamping or other processing of the parts for the electrical applications, but develops hardening by heat
35 treatment after the processing, having high strength and wear resistance as well as excellent magnetic properties.

The present invention has been made to solve the

above problems of providing the electrical steel sheet including Cu followed by proper heat treatment so as to include a metal phase fine Cu and obtaining high strength, high wear resistant electrical steel sheet without inviting a deterioration of magnetic properties or productivity that are accompanied by conventional high strength electrical steel sheet. The gist of the present inventions are as follows.

(1) A high strength electrical steel sheet and a processed part of the same characterized by containing, by mass%, C: 0.06% or less, Si: 0.2 to 6.5%, Mn: 0.05 to 3.0%, P: 0.30% or less, S or Se: 0.040% or less, Al: 2.50% or less, Cu: 0.6 to 8.0%, N: 0.0400% or less, and a balance of Fe and unavoidable impurities and containing in the steel a metal phase comprised of Cu of a diameter of 0.1 μm or less.

(2) A high strength electrical steel sheet and a processed part of the same as set forth in (1), characterized by further containing, by mass%, one or more of Nb: 8% or less, Ti: 1.0% or less, B: 0.010% or less, Ni: 5% or less, and Cr: 15.0% or less.

(3) A high strength electrical steel sheet and a processed part of the same as set forth in (1) or (2), characterized by further containing, by mass%, one or more of Bi, Mo, W, Sn, Sb, Mg, Ca, Ce, La, and Co in a total of 0.5% or less.

(4) A high strength electrical steel sheet and a processed part of the same as set forth in any one of (1) to (3), wherein the number density of the metal phase comprised of Cu present in said steel is $20/\mu\text{m}^3$ or more.

(5) A high strength electrical steel sheet and a processed part of the same as set forth in any one of (1) to (4), wherein said steel sheet or the part has an average crystal grain size of 30 to 300 μm .

(6) A high strength electrical steel sheet and a processed part of the same as set forth in any one of (1)

to (5), wherein the steel sheet or the part has a worked structure remaining in it.

5 (7) A high strength electrical steel sheet and a processed part of the same as set forth in any one of (1) to (6), characterized in that the steel sheet or the part contains Nb carbide or nitride.

10 (8) A method of production of a high-strength electrical steel sheet and a processed part of the same as set forth in any one of (1) to (7), wherein the sheet or the part is held at a temperature range of 300°C to 720°C for 5 seconds or more for heat treatment in the production process.

15 (9) A method of production of a high strength electrical steel sheet and a processed part of the same as set forth in (8), characterized by, as said heat treatment, holding at a temperature range of 300°C to 720°C for 5 seconds or more in a cooling process from a temperature range of 750°C or more in a final heat treatment process.

20 (10) A method of production of a high strength electrical steel sheet and a processed part of the same as set forth in (8) or (9), characterized by, after the heat treatment, holding in a temperature range over 800°C for 20 seconds or more.

25 (11) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7), characterized wherein the part is heat treated after processing for shaping so that the metal phase comprised mainly of Cu present in the processed part has a number
30 density of $20/\mu\text{m}^3$ or more.

(12) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7) and (11), characterized wherein the part is heat treated after processing for shaping so that the metal phase
35 comprised mainly of Cu present in the part has an average size of 0.1 μm or less.

(13) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7) and (11) and (12), characterized wherein the part is heat treated after processing for shaping so that the part has
5 an average crystal grains size of 3 to 300 μm .

(14) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7) and (11) to (13), characterized wherein the part is heat treated after processing for shaping so that the number
10 density of the metal phase comprised mainly of Cu with a size of 0.1 μm or less in the processed part is increased by 10-fold or more.

(15) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7) and (11) to (14), wherein the part is heat treated after
15 processing for shaping so that the tensile strength is increased by 30 MPa or more.

(16) A processed part of a high strength electrical steel sheet as set forth in any one of (1) to (7) and (11) to (15), wherein the part is heat treated after
20 processing for shaping so that the hardness is increased by 1.1-fold or more.

(17) A method of production of a high strength electrical steel sheet as set forth in any one of (11) to (16), characterized by making the residence time in the
25 temperature range of 450°C to 700°C in the cooling process from a temperature range of 750°C or more after the hot rolling process before cold rolling 300 seconds or less, then cold rolling without holding in a temperature range
30 over 750°C so as to keep the steel soft before processing for shaping and harden it by heat treatment after the processing for shaping.

(18) A method of production of a high strength electrical steel sheet as set forth in (17),
35 characterized by holding at 750°C or more in a final heat treatment process after hot rolling and cold rolling,

then making the residence time in the temperature range of 450°C to 700°C in the cooling process from the temperature range of 750°C or more 60 seconds or less, then not holding in a temperature range over 750°C so as to keep the steel soft before processing for shaping and harden it by heat treatment after the processing for shaping.

(19) A method of production of a processed part of a high strength electrical steel sheet characterized by processing for shaping the electrical steel sheet as set forth in any of (1) to (7) and (11) to (16) or an electrical steel sheet produced by a method as set forth in any of (17) and (18), then holding in a temperature range of 300°C to 720°C for 5 seconds or more, then not holding in a temperature range over 700°C for 20 seconds or more to obtain the processed part.

(20) A method of production of a processed part of a high strength electrical steel sheet as set forth in (19), characterized by, as said heat treatment method, making an average cooling rate of a cooling process from the heat treatment temperature to 700°C in heat treatment after processing the steel sheet to an electrical part 10°C/seconds or more, holding in a temperature range of 300°C to 720°C for 5 seconds or more, then not holding in a temperature range over 700°C for 20 seconds or more.

BRIEF DESCRIPTION OF THE DRAWINGS

FIG. 1 is a schematic drawing of the relationship between the Si content and tensile strength of the electrical steel sheet of the present invention.

FIG. 2 is a schematic drawing of the relationship between the tensile strength and core loss of the present invention.

BEST MODE FOR WORKING THE INVENTION

First, the composition of the high strength electrical steel sheet according to the present invention will be described in detail.

Carbon degrades magnetic properties of the sheet and the amount is 0.06% or less. Carbon is effective from the viewpoint of increasing the strength, in particular raising the yield stress, improving the strength in hot condition and creep strength, and improving the fatigue properties in hot condition. It is also effective in improving the texture by suppressing the development of {111} orientation not desirable for magnetic properties and accelerating development in the preferable {110}, {100}, {114} etc. From this viewpoint, the amount is preferably 0.04% or less, more preferably 0.0031 to 0.0301%, more preferably 0.0051 to 0.0221%, more preferably 0.0071 to 0.0181%, more preferably 0.0081 to 0.0151%. If within the range of the present invention, it is possible to suppress the magnetic aging by using gradual cooling, holding at a low temperature, or other heat histories to an extent not so much problem.

On the other hand, when the requirements regarding magnetic aging are extremely severe, it is possible to include a higher amount of C up to the slab production stage from the viewpoint of the deoxidation efficiency and to reduce the C to 0.0040% or less by decarburizing annealing after being made into a coil. In this case, from the viewpoint of production costs, it is advantageous to reduce the amount of C by a degasification facility at the molten steel stage. If 0.0020% or less, there is a remarkable effect of suppression of magnetic aging. To increase the strength, when not using carbides or other nonmetal precipitates, C content is preferably 0.0015% or less, and more preferably 0.0010% or less.

Silicon increases volume resistivity of the steel reducing the eddy current to reduce the core loss and increases the tensile strength, but if the amount added is less than 0.2%, that effect is small. Increasing the Si content degrades less magnetic properties and in particular it is possible to reduce the core loss and to

increase the strength, so Si content is preferably in an amount of 1.0% or more, more preferably 2.0% or more, in the steel. On the other hand, if Si is over 6.5%, the steel is embrittled and further the magnetic flux density of the product sheet is degraded, so the amount is 6.5% or less, preferably 3.5% or less. To further reduce the concern over the brittleness, 3.2% or less Si is preferable. If Si is 2.8% or less, there is no longer a need to consider the brittleness although it is related with the amounts of other elements.

Manganese may be purposely added to improve the steel in strength, but is not particularly required for this purpose in the steel of the present invention utilizing a fine metal phase as the main means for increasing the strength. This is added to for the purpose of increasing the volume resistivity or coarsening the sulfides to promote crystal grain growth reducing the core loss, but since excessive addition reduces the magnetic flux density, the amount is 0.05 to 3.0%. Preferably, the amount is 0.5% to 1.2%.

Phosphorus is an element with a remarkable effect of improving the tensile strength, but like with the above mentioned Mn, in the steel of the present invention, the addition is not necessary. If P amount is over 0.30%, it causes so severe brittleness that hot rolling, cold rolling, or other processing in an industrial scale is difficult, so the upper limit of P is 0.30%.

Sulfur easily combines with Cu, the essential element in the steel of the present invention, to form a Cu sulfides and thereby has an adverse effect on the formation of the metal phase mainly composed of Cu which is important in the present invention resulting in degrading the strengthening efficiency in some cases, so care is required when including it in large amounts. Further, depending on the heat treatment conditions, it is also possible to purposely form fine Cu sulfides and promote an increase in strength. Thus produced sulfides

sometimes degrades the magnetic properties, in particular the core loss. In particular, in the case of non-oriented electrical steel sheet, a low S content is preferable, so Cu content is limited to 0.040% or less. Preferably, it is 0.020% or less, more preferably 0.010% or less. Selenium also has substantially the same effect as S.

Aluminum is usually added as a deoxidizing agent although it is also possible to add less Al and use Si for deoxidization. In particular, in the case of a non-oriented electrical steel sheet, AlN is not formed in Si-deoxidized steel with an Al content of about 0.005% or less, so there is also the effect of reducing the core loss. On the other hand, it is possible to purposely add it to promote coarsening AlN and increase the volume resistivity to reduce the core loss. However, if Al content is over 2.50%, embrittlement becomes a problem, thus the content is 2.50% or less.

Cu is an essential element in the present invention. The range for forming a metal phase mainly composed of Cu in the steel sheet to increase the strength not having a adverse effect on the magnetic properties is limited to 0.6 to 8.0%. More preferably, it is 0.8 to 6.0%. If the Cu is low in content, the effect of increasing the strength becomes small, the heat treatment conditions for obtaining the effect of increasing strength are limited to a narrow range, and the flexibility of control of the production conditions and adjustment of production becomes smaller. Further, if the Cu is high in content, the effect on the magnetic properties becomes larger resulting in remarkably increased core loss, and increased cracking or flaws in the steel sheet at the time of hot rolling.

In particular, an amount of Cu over the limit of solid solution in the steel contributes to increased strength as solute Cu, but the strengthening efficiency becomes poor compared with the Cu metal phase of the main object of the present invention. Further, excessive Cu

forms a metal phase in the steel in a not preferable process depending on the heat history. For example, Cu forms a relatively coarse Cu metal phase in a high temperature such as during hot rolling, which acts
5 unpreferably on the formation of the subsequent fine metal phase or has a detrimental effect on the magnetic properties in some cases. The particularly preferable range of Cu is 1.0 to 5.0%. More preferably, it is 1.5 to 4.0%, more preferably 2.0 to 3.5%.

10 Nitrogen, like C, degrades the magnetic properties, and thus the amount is 0.0400% or less.

In Si-deoxidized steel with Al of about 0.005% or less, like C, it is an element effective from the viewpoint of improving the texture in addition to
15 increasing the strength, in particular increasing the yield stress, increasing the high temperature strength and creep strength, and increasing the fatigue properties. From this viewpoint, it is preferably 0.0031 to 0.0301%, more preferably 0.0051 to 0.0221%, more
20 preferably 0.0071 to 0.0181%, more preferably 0.0081 to 0.0151%. However, when Al is about 0.010% or more, if a large amount of N is included, fine AlN is formed and the magnetic properties are remarkably degraded, so this must be avoided.

25 In Al-deoxidized steel, N content should be 0.0040% or less. In the present invention where no strengthening due to nitrides is expected, the lower the better. If N is 0.0027% or less, there is a remarkable effect of suppression of magnetic aging or degradation of
30 properties due to AlN in the Al-containing steel. Therefore, N is more preferably 0.0022%, more preferably 0.0015% or less.

Almost all elements utilized in the past for increasing the strength in high strength electrical steel
35 sheet not only are problematic in terms of the cost of addition, but also have some detrimental effect on the magnetic properties, so, in the present invention, they

do not really have to be added for the purpose of increasing the strength. When they are purposely added as strengthening elements, due to the relation with rising costs and degradation of magnetic properties, one type or
5 more of Nb, Ti, B, Ni, and Cr may be added but the amounts added are Nb: 8% or less, preferably 0.02% or less, Ti: 1.0% or less, preferably 0.010% or less, B: 0.010% or less, Ni: 5.0% or less, and Cr: 15% or less, preferably 10.0% or less.

10 In particular, Ni is known to be effective to prevent surface roughening by Cu, the essential element in the steel of the present invention, at hot rolling (Cu scab) and may be purposely added for this purpose as well. Boron segregates at the crystal grain boundaries
15 and has the effect of suppressing embrittlement due to grain boundary segregation of P, but in the steel of the present invention, embrittlement does not become a particular problem like in the conventional mainly solution strengthened high strength electrical steel
20 sheet, so addition for this purposes is not important. Rather, it may be added for the purpose of improving the magnetic flux density due to the effect of the solute B on the texture. If B is over 0.010%, remarkable embrittlement results, so the upper limit of B is 0.010%.

25 Niobium and Titanium form fine precipitates of carbides, nitrides, or sulfides in the steel sheet and are elements effective for increasing the strength, but simultaneously cause remarkable degradation of the magnetic properties, in particular the core loss. In the
30 steel of the present invention not utilizing fine carbides, nitrides, etc. as main means for increasing the strength, these are rather harmful elements. For this reason, the upper limit for Nb is 8% or less, preferably 0.02% or less, while that for Ti is 1.0% or less,
35 preferably 0.010%. With both, the limit is more preferably 0.0050% or less, and further preferably 0.0030% or less, whereby it is possible to obtain a good

core loss.

Nickel is known to be effective for the prevention of surface roughening at hot rolling due to Cu (Cu scab), the essential element in the steel of the present invention, and can be purposely added together with this purpose. Further, it has a relatively small detrimental effect on the magnetic properties and is deemed effective for increasing the strength as well, so is an element often used in high strength electrical steel sheet. For the purpose of preventing Cu roughening, it is added in an amount of roughly about 1/8 to 1/2 of the amount of Cu. Further, in the steel of the present invention increased in strength utilizing the Cu metal phase, by including Ni as well, the dispersion of the metal Cu phase also becomes extremely preferable for suppressing degradation of the magnetic properties and increasing the strength. The reason for this is not clear, but is expected to be for example the effect of the solid solution of the Ni in the metal Cu phase or the formation of a metal phase related with Ni and Cu. Further, it is also effective for improvement of the corrosion resistance, but considering the cost of addition and the detrimental effects on the magnetic properties, the upper limit is preferably 5% and further preferably 2.5%.

Chromium is an element added to improve the corrosion resistance or to improve the magnetic properties in high frequency, but again considering the cost of addition and the detrimental effects on the magnetic properties, preferably the upper limit is 15% and more particularly 10.0%.

Further, regarding the other minor elements, not only the amounts unavoidably contained due to the ore, scrap, etc., but the amounts added for various purposes do not impair the effects of the present invention in any way. The unavoidable contents of these elements are usually about 0.005% or less each, but addition in amounts of 0.01% or more is possible for various

purposes. In this case as well, considering the balance of the cost and magnetic properties, the steel may contain one or more of Bi, Mo, W, Sn, Sb, Mg, Ca, Ce, La, Co, and other rare earth elements in a total of 0.5% or less.

The steel containing these compositions is produced in the same way as an conventional electrical steel sheet wherein it is melted in a converter, continuously cast into a slab, hot rolled, hot band annealed, cold rolled, final annealed and so on. In addition to these processes, formation of an insulating film or a decarburization process etc. do not impair the effect of the present invention in any way. Further, unusual process such as production of thin strip by rapid solidification or continuous casting of thin slabs omitting the hot rolling process have no problems.

To form the specific metal phase characteristic of the present invention in the steel sheet, it is effective to go through the following heat history. In the process of production of a product sheet, the sheet is held at a temperature range of 300°C to 720°C for 5 seconds or more. The temperature range is preferably 300 to 650°C, more preferably 350 to 600°C, more preferably 400 to 550°C, more preferably 420 to 500°C. The holding time is related to the holding temperature. Preferably, the lower the temperature, the longer the time. On the other hand, holding at a high temperature for a long time is not preferable. Preferably, it is 650°C or so for 1 minute to 5 hours, at 550°C or so for 3 minutes to 20 hours, and at 450°C or so for 10 minutes or more.

Further, after this heat treatment, it is preferable to avoid a process holding the heat treated steel sheet at a temperature range over 800°C for 20 seconds or more.

Through the above process, a Cu metal phase of characteristic composition, size, and number density is efficiently formed, the magnetic properties are not

impaired much at all, and the strength can be increased. On the other hand, through ordinary heat treatment conditions not aiming at formation of such a metal phase, the majority of the Cu added forms solute Cu or Cu sulfides which are low in strengthening ability and large in effect of degradation of magnetic properties or a relatively coarse Cu metal phase which is small in strengthening ability and large in detrimental effect on the magnetic properties although it is a Cu metal phase,.

After this heat treatment process, the steel is increased in strength, so performing this heat treatment process after the rolling process and performing it simultaneously with the recrystallization annealing or other heat treatment required for other purposes is advantageous from the viewpoint of productivity. That is, holding in the temperature range of 300°C to 720°C for 5 seconds or more in the final heat treatment process after the cold rolling in the case of cold rolled electrical steel sheet or in the cooling process from the temperature range of 750°C or more in the final heat treatment process after hot rolling in the case of hot rolled electrical steel sheet is preferable. The effect of this heat treatment depends on the steel components, in particular the amounts of Cu, Ni, etc., but some sort of effect may occur even with a heat history of a cooling rate of such as air cooling after recrystallization annealing.

Further, depending on the targeted properties etc., heat treatment is sometimes further added, but in this case, it is preferable not to hold the steel in a temperature range over 800°C for 20 seconds or more. When the temperature and time exceed this in the heat treatment, the formed Cu metal phase resolidifies or conversely aggregates to form a coarse metal phase in some cases. In particular, when the metal phase becomes coarsened, the core loss remarkably deteriorates.

Since the present invention does not utilize strengthening by refinement of crystal structure, there is little degradation of the strength even if performing SRA (stress relief annealing) for relieving the stress introduced into the material and growing the crystal grains to restore and improve the magnetic properties when stamping steel sheet and processing it to motor parts etc. or some other heat treatment performed for other purposes.

Further, it is important that the specific metal phase characteristic of the present invention go through the following heat history for formation in the steel sheet after processing to an electrical part. This is to control the holding time in the 300°C to 720°C temperature range and the subsequent heat history in the process of producing the product sheet and the heat treatment process after being processed to an electrical part.

That is, as the heat treatment given to the steel sheet up until the final processing, i.e. punching and fabricating process for utilizing the electrical steel sheet as an electrical part, it is preferable to make the residence time in the temperature range of 450°C to 700°C in the cooling process from the temperature range of 750°C or more to 300 seconds or less for the heat history before the cold rolling after the final hot rolling and 60 seconds or less for the annealing process after cold rolling respectively, and then not holding at a temperature range over 750°C.

Further, the hardening is performed after the final processing of the electrical steel sheet, i.e. the punching and fabricating process for utilization of the electrical steel sheet for an electrical part. It can be achieved by a heat treatment comprising holding the steel at a temperature range of 300°C to 720°C for 5 seconds or more, then not holding at a temperature range over 700°C for 20 seconds or more. When this heat treatment is

performed in the cooling process after a heat treatment at a higher temperature, the average cooling rate of the cooling process down to 700°C before holding in the temperature range of 450°C to 700°C is preferably

5 10°C/second or more, more preferably the average cooling rate of the cooling process down to 650°C before holding in the temperature range of 500°C to 650°C is 10°C/second or more. Performing this heat treatment in the cooling process such as so-called stress relief annealing process

10 performed for the purpose of relieving the stress introduced against intent in the material when processing, or the heat treatment for burning off the oil adhering to the steel sheet at processing is preferable from the viewpoint of the productivity. The maximum peak

15 temperature of 700°C or more before holding in the temperature range of 300°C to 720°C and the holding time in that temperature range can be determined from only the viewpoint of stress relief and crystal grain growth and do not have any influence on the effects of the present

20 invention.

The holding temperature range in the temperature range of 300°C to 720°C for hardening is preferably 300 to 650°C, more preferably 350 to 600°C, further preferably 400 to 550°C, and further preferably 420 to 500°C. The

25 holding time is related to the holding temperature. It is preferable that the lower the temperature, the longer the time. On the other hand, holding at a high temperature for a long time is not preferable. Preferably, if holding at around 650°C for 1 minute to 5 hours, at around 550°C

30 for 3 minutes to 20 hours, and at 450°C or so for 10 minutes or more, a sufficient hardening effect can be obtained.

Passing through this process results in efficient formation of the metal phase characteristic in

35 composition, size, and number density in a preferable

process and enables hardening without impairing the magnetic properties much at all. According to the present invention, the steel can be increased in tensile strength by 30 MPa or more or in hardness by 10% or more by heat treatment for hardening. If the increment in strength or hardness is less than this, probably the steel is already hardened before the heat treatment or the strengthening ability by heat treatment is originally not provided.

If the sheet is already hardened before heat treatment, punching into the motor part etc. is performed on a hard material, which is not preferable from the viewpoint of the wear of the dies. Further, when not hardened even with heat treatment, the strength during use as a motor becomes insufficient and the object of the present invention is not achieved. To obtain a more preferable effect, preferably the increase in tensile strength due to the heat treatment is 60 MPa or more and the increase in hardness is 20% or more, more preferably the increase in tensile strength is 100 MPa or more and the increase in hardness is 30% or more, further preferably the increase in tensile strength is 150 MPa or more and the increase in hardness is 40% or more, and further preferably the increase in tensile strength is 200 MPa or more and the increase in hardness is 50% or more.

On the other hand, when passing through normal heat treatment conditions not aiming at formation of a metal phase as controlled in the present invention, depending on the steel composition, sometimes the effect of the formation of a metal phase may be detected, but the majority of the added Cu is present as solute Cu or Cu sulfides or a coarse metal phase over a size of 0.1 μm which are low in strengthening ability and large in effect of degradation of the magnetic properties.

The metal phase formed in the above way is mainly comprised of Cu. This can be identified by the

diffraction parameters of an electron microscope etc. or an attached X-ray analysis apparatus etc. Of course, it may also be identified by chemical analysis or another method. In the present invention, this metal phase mainly
5 comprised of Cu has a size 0.1 μm or less, more preferably 0.01 μm or less. If the size is more than this, the efficiency of increasing the strength falls. In this condition, not only a large amount of metal phase becomes necessary, but also the detrimental effect on the
10 magnetic properties becomes greater. From the viewpoint of increasing the strength and the magnetic properties, the diameter is preferably 0.008 μm or less, further 0.005 μm or less, and more preferably 0.002 μm or less. Note that if less than 0.001 μm , the size becomes too
15 fine and quantification of the metal phase size and amount of metal phase would become difficult even by the current highest precision analysis equipment, but identification by X-ray analysis equipment and indirect explanation of presence by mechanical properties,
20 hardness, etc. would be possible. The present invention is limited to an electrical steel sheet which contains a considerable amount of Cu and clearly hardens by appropriate heat treatment explained in the present invention. Needless to say, while the present invention
25 describes a "Cu metal phase", this does not limit its form or type .

The number density of the Cu metal phase is limited in the possible range in the relation between the Cu content and size of the metal phase, but preferably is
30 0.2/ μm^3 or more, 1/ μm^3 or more, 5/ μm^3 or more, more preferably 20/ μm^3 or more, more preferably 50/ μm^3 or more, 100/ μm^3 or more, or 200/ μm^3 or more, more preferably 500/ μm^3 or more, 1,000/ μm^3 or more, 2,000/ μm^3 or more. If so, this is extremely effective in the point of
35 increasing the strength. More preferably, it is 5,000/ μm^3

or more, $10,000/\mu\text{m}^3$ or more, $20,000/\mu\text{m}^3$, more preferably $200,000/\mu\text{m}^3$, more preferably $2,000,000/\mu\text{m}^3$.

Control of the metal phase size and number density is extremely important from the viewpoint of achieving both increased strength and holding the magnetic properties. The reason is that these are not only effective on the strength and magnetic properties, but also the way of change of strength and magnetic properties differs when these are changed. That is, it is necessary to control these to the range where the effect of increasing the strength is high and the efficiency of degradation of the magnetic properties is low. For this reason, it is effective to suitably control the temperature and time in the above-mentioned temperature range of 300 to 720°C and the cooling rate immediately before entering this temperature range. The effect is, under normal conditions, like as the formation of general precipitates, if the cooling rate is higher and the temperature is lower, the precipitates are finer and the metal phase density is higher. A long time leads to a coarser size.

Further, in the present invention, since refinement of the crystal structure is not utilized as the main means for increasing the strength, the crystal grain size can be adjusted to the optimal range from the viewpoint of the magnetic properties. The size and density of the metal phase mainly comprised of Cu contributing to the increased strength can be controlled not only by the components, but also mainly by the above-mentioned heat treatment at 720°C or less, so the crystal grain size can be independently controlled from the strength by for example the maximum peak temperature of the recrystallization annealing and the holding time in this temperature range before the heat treatment. Normally, it is controlled to 3 μm to 300 μm by heat treatment at 800°C to 1100°C or so for 20 seconds to 5 minutes or so. More

preferably, it is 8 μm to 200 μm . In general, when the frequency of the magnetization current at the time of use of the steel sheet is high, the crystal grains are preferably fine.

5 The present invention has properties completely different from the materials developed in the past for magnetic steel sheet. FIGs. 1 and 2 show the characteristics of the present invention from the viewpoint of the compositions, strength, and magnetic
10 properties of the electrical steel sheet. As shown in FIG. 1, usually electrical steel sheet is produced selectively for magnetic properties mainly by the Si content. From the viewpoint of the magnetic properties, Si is usually added to increase the electrical resistance
15 of the material and to reduce the core loss, but since Si also has a strong solid solution strengthening effect, the strength is also increased in high Si, i.e. high grade, materials. However, if the amount of Si is over 3% or the combination of Si, Al, Mn and other strengthening
20 elements exceeds 6.5%, the rollability remarkably degrades and production of the steel sheet becomes difficult in conventional production process.

 As the means for avoiding rolling, the method of directly obtaining a thin film from molten state steel by
25 rapid solidification has been proposed, but there are limits to practical use of this from the viewpoint of cost and properties. For this reason, a high strength material equivalent to 3%Si steel or more is strengthened by the precipitates mainly comprised of carbonitrides
30 accompanying addition of Nb etc. and the refinement of the crystal structure also involving low temperature annealing. However, such carbonitrides or a fine crystal structure is not preferable from the viewpoint of the magnetic properties, in particular the core loss as shown
35 in FIG. 2, wherein a great increase in the core loss is unavoidable. In this invention, however, in so far as the magnetic properties are not remarkably impaired, the

steel sheet of the present invention may contain these carbonitrides or may have a residual deformation texture in part. In other words, it is possible to use the effect of hardening by the Cu metal phase according to the present invention in combination with the conventional high strength steel using carbonitrides or high strength steel using the deformation texture so as to further increase the strength. In particular, the steel of the present invention containing high amount of Cu may leave a residual deformation texture under low temperature annealing conditions because of highrecrystallization temperature depending on the components or heat history.

The present invention disperses a metal phase in a steel sheet so as to increase the strength, which is different from the conventional high strength steels. This metal phase can be controlled independently of the crystal grain size, that is, formation of the metal phase can be controlled in a lower temperature range of about 300 to 720°C which is different from the temperature range where crystal grain growth occurs, that is, about 750°C or more. Therefore the present invention has a greater flexibility from the viewpoint of controlling the strength and magnetic properties independently and thus, as shown in FIG. 2, the strength can be increased without degrading the magnetic properties very much.

Further, as shown in FIG. 1, by applying this technology to low Si steel, it becomes possible to obtain a material with a higher magnetic flux density than conventional one. This is considered to be as follows: because most of the solid solution strengthening elements including typically used Si, Al, Mn reduce the saturation magnetic flux density of the steel, lowered magnetic flux density in a specific magnetic field is unavoidable, but in the present invention, the Cu metal phase used for increasing the strength has an extremely small effect in reducing the saturation magnetic flux density. Furthermore, it is considered that the Cu metal phase

does not become a barrier for magnetic domain wall motion compared with other precipitates such as carbonitrides. This is effective for improvement of the magnetic properties especially in a low magnetic field.

5 Note that the the effects of the present invention are not affected by the existence and type of surface coating normally formed on the surface of the electrical steel sheet or by the production process, so that it can be applied to non-oriented or grain-oriented electrical
10 steel sheet.

The applications are also not particularly limited. The sheet can be applied not only to the rotors of motors used in home electric appliances or automobiles etc., but also to all other applications where both strength and
15 magnetic properties are required.

EXAMPLES

(Example 1)

The steel compositions shown in Table 1 was made into a 250 mm thick slab and final sheets were produced by
20 basically following process. The basic process conditions were a slab heating temperature of 1100°C, a final hot band thickness of 2.0 mm, a coiling temperature of 500°C in hot rolling, a final sheet thickness of 0.5 mm in the cold rolling, and a recrystallization annealing
25 temperature of 850°C. Each product sheet was measured for mechanical properties using a JIS No. 5 test piece and for magnetic flux density B_{10} and core loss $W_{10/400}$ using a 55 mm square SST test. The mechanical properties and magnetic properties were calculated as the average of the
30 values of the rolling direction and transverse direction of the coil. The results are shown in Table 2 (continuation of Table 1).

As clear from the results shown in Table 2, the samples produced by the conditions of the present
35 invention are good in rollability at the cold rolling process, hard, and superior in magnetic properties.

Table 1

No.	Steel compositions (mass%)										Precipitation process		High temperature treatment after precipitation	
	C	Si	Mn	P	S	Al	N	Cu	Others	Temp- erature (°C)	Holding time (min)	Temp- erature (°C)	Holding time (sec)	
1	0.0020	1.1	0.32	0.021	0.0030	0.002	0.0024	0.006*	-	-	-	-	-	
2	0.0049	1.1	0.33	0.121	0.0023	0.003	0.0039	0.007*	Ni:1.5 Nb:0.03	-	-	-	-	
3	0.0008	1.2	0.13	0.016	0.0023	0.001	0.0023	0.93	-	650	60	-	-	
4	0.0015	1.3	0.32	0.015	0.0010	0.002	0.0017	1.41	-	700	120	-	-	
5	0.0016	1.1	0.30	0.020	0.0026	0.005	0.0015	1.14	-	550	10	-	-	
6	0.0022	1.2	0.44	0.021	0.0260	0.002	0.0014	1.40	Ni:1.3	500	60	-	-	
7	0.0022	1.2	0.44	0.021	0.0260	0.002	0.0059	1.40	Ni:1.3	500	60	-	-	
8	0.0022	1.2	0.44	0.021	0.0260	0.002	0.0165	1.40	Ni:1.3	500	60	-	-	
9	0.0022	1.2	0.44	0.021	0.0053	0.002	0.0014	1.83	Ti:0.03	450	120	-	-	
10	0.0022	1.2	0.44	0.021	0.0021	0.002	0.0014	2.41	-	400	150	-	-	
11	0.0022	1.2	0.44	0.021	0.0021	0.002	0.0105	2.41	-	400	150	-	-	
12	0.0022	1.2	0.44	0.021	0.0021	0.002	0.0211	2.41	-	400	150	-	-	
13	0.0027	1.1	0.80	0.020	0.0005	0.003	0.0020	2.74	-	300	1000	-	-	
14	0.0091	1.1	0.80	0.020	0.0005	0.003	0.0020	2.74	-	300	1000	-	-	
15	0.0018	1.3	0.30	0.016	0.0013	0.002	0.0022	8.53*	-	500	120	-	-	

*Constitutions outside range of claims of present invention

Table 2

No.	Metal phase composed of Cu		Crystal grain average size (μm)	Mechanical properties				Magnetic properties		Cold roll-ability	Evaluation
	Average size (μm)	Number density ($/\mu\text{m}^3$)		Hardness Hv	YP (MPa)	TS (MPa)	EI (%)	B ₁₀ (T)	W _{10/100} (W/kg)		
1	-	-	100	140	276	475	35	1.63	35.0	VG	D
2	-	-	25	240	518	682	14	1.30	52.9	F	D
3	0.35*	0.2	120	216	448	573	32	1.62	37.3	VG	D
4	0.22*	2.5	160	250	533	623	25	1.60	34.8	VG	D
5	0.008	30	180	245	545	690	28	1.60	32.5	VG	C
6	0.005	200	90	265	588	756	19	1.62	34.1	VG	B
7	0.005	200	90	270	622	765	17	1.64	33.1	VG	B
8	0.005	200	85	276	673	769	14	1.64	33.5	VG	B
9	0.002	2000	70	288	576	699	25	1.58	36.1	VG	B
10	0.001	400	120	276	622	789	26	1.60	29.7	VG	A
11	0.001	400	115	298	725	792	20	1.63	30.1	VG	A
12	0.001	400	120	302	755	815	22	1.61	27.6	VG	A
13	0.001	>10000	100	310	788	923	20	1.57	37.2	VG	A
14	0.001	>10000	100	325	882	944	15	1.59	35.9	VG	A
15	0.35*	2.3	70	249	547	704	25	1.40	48.8	G	D

Evaluation

A: Developed steel (very good)

B: Developed steel (good)

C: Developed steel (slightly good)

D: Comparative steel

Cold rollability

VG: Very good (no problem at all)

G: Good (fine adjustment required, but no problem)

F: Fair (sheet runnable if adjusting conditions)

P: Poor (large danger of sheet breakage)

(Example 2)

The steel compositions shown in Table 3 was made into a 250 mm thick slab and final sheets were produced by basically following process. The basic process conditions were a slab heating temperature of 1100°C, a final hot band thickness of 2.0 mm, a coiling temperature of 700°C in the hot rolling, a hot band annealing of 980°C temperature for 30 seconds, a final sheet thickness of 0.2 mm in the cold rolling, and a recrystallization annealing of 1000°C. Each product sheet was measured for mechanical properties using a JIS No. 5 test piece and for magnetic flux density B_{50} and core loss $W_{15/50}$ using a 55 mm square SST test. The mechanical properties and magnetic properties were calculated as the average of the values of the rolling direction and transverse direction of the coil. The results are shown in Table 4 (continuation of Table 3).

As clear from the results shown in Table 4, the samples produced by the conditions of the present invention are good in rollability at the cold rolling process, hard, and superior in magnetic properties.

Table 3

No.	Steel compositions (mass%)											Precipitation process			High temperature treatment after precipitation	
	C	Si	Mn	P	S	Al	N	Cu	Other	Temp- erature (°C)	Holding time (min)	Temp- erature (°C)	Holding time (sec)			
16	0.0021	3.1	0.18	0.014	0.0005	0.51	0.0018	0.008*	-	-	-	-	-			
17	0.0088	3.1	0.20	0.014	0.0017	0.56	0.0012	0.026*	Ni:2.5 Nb:0.03	-	-	-	-			
18	0.0009	2.8	0.19	0.005	0.0024	0.57	0.0011	0.97	-	750	120	-	-			
19	0.0021	3.1	0.20	0.007	0.0012	0.52	0.0006	1.37	-	700	0.5	-	-			
20	0.0021	3.1	0.21	0.005	0.0049	0.54	0.0011	1.66	-	600	2	-	-			
21	0.0011	2.9	0.19	0.014	0.0068	0.56	0.0008	1.52	Cr:4.5	500	5	1050	30			
22	0.0021	2.5	0.16	0.004	0.0023	0.52	0.0023	3.31	-	450	120	-	-			
23	0.0051	2.5	0.16	0.004	0.0023	0.52	0.0023	3.31	-	450	120	-	-			
24	0.0112	2.5	0.16	0.004	0.0023	0.52	0.0023	3.31	-	450	120	-	-			
25	0.0145	2.5	0.16	0.004	0.0023	0.52	0.0023	3.31	-	450	120	-	-			
26	0.0205	2.5	0.16	0.004	0.0023	0.52	0.0023	3.31	-	450	120	-	-			
27	0.0013	3.1	0.24	0.012	0.0130	0.88	0.0010	2.85	Ni:1.4	400	600	-	-			
28	0.0255	3.1	0.24	0.012	0.0130	0.88	0.0010	2.85	Ni:1.4	400	600	-	-			
29	0.0009	2.8	0.21	0.008	0.0012	0.55	0.0013	5.12	Ni:2.0	350	1800	1000	40			
30	0.0015	3.1	0.18	0.009	0.0023	0.55	0.0011	8.72*	-	450	120	-	-			

*Constitution outside range of claims of present invention

Table 4

No.	Metal phase composed of Cu		Crystal grain average size (μm)	Mechanical properties				Magnetic properties		Cold rollability	Evaluation
	Average size (μm)	Number density ($/\mu\text{m}^3$)		Hardness Hv	Yp (MPa)	TS (MPa)	EI (%)	B ₉₀ (T)	W _{15/50} (W/kg)		
16	-	-	120	176	390	530	17	1.68	2.4	F	D
17	-	-	20	320	804	833	14	1.64	10.6	P	D
18	0.55*	0.2	210	315	833	881	13	1.69	4.4	G	D
19	0.01	1.2	130	315	789	905	13	1.67	3.2	F	C
20	0.004	2000	110	309	738	938	14	1.65	2.5	F	B
21	0.008	2000	95	397	944	1119	11	1.69	2.5	G	B
22	0.002	>10000	210	355	790	980	14	1.70	2.4	VG	A
23	0.002	>10000	200	360	820	985	12	1.72	2.4	VG	A
24	0.002	>10000	190	363	890	1023	12	1.73	2.3	VG	A
25	0.002	>10000	190	372	874	1030	11	1.71	2.3	VG	A
26	0.002	>10000	180	380	886	1032	12	1.69	2.4	VG	A
27	0.001	5000	180	379	1055	1205	10	1.71	2.1	VG	A
28	0.001	5000	180	399	1150	1210	8	1.73	2.1	VG	A
29	0.003	600	130	325	856	898	16	1.67	2.7	F	B
30	0.25*	200	60	267	690	793	20	1.53	3.8	P	D

Evaluation

Cold rollability

A: Developed steel (very good)

VG: Very good (no problem at all)

B: Developed steel (good)

G: Good (fine adjustment required, but no problem)

C: Developed steel (slightly good)

F: Fair (sheet runnable if adjusting conditions)

D: Comparative steel

P: Poor (large danger of sheet breakage)

(Example 3)

The steel compositions shown in Table 5 was made into a 250 mm thick slab and final sheets were produced by basically following process. The basic process conditions were a slab heating temperature of 1100°C, a final hot band thickness of 2.0 mm, a coiling temperature of 300°C or less in hot rolling, a final sheet thickness of 0.2 mm in cold rolling, and a recrystallization annealing temperature of the temperature or more at which the recrystallization occurred. After this, to simulate precipitation heat treatment after punching, a heat treatment at about 750°C was employed to control texture and precipitation of the metal phase. When the heat treatment simulates stress relief annealing, the precipitation heat treatment was employed in the cooling process after a heat treatment at 750°C for 2 hours. Each sheet before and after the heat treatments was measured for mechanical properties using a JIS No. 5 test piece and for magnetic flux density B_{10} and core loss $W_{10/400}$ using a 55 mm square SST test. The mechanical properties and magnetic properties were calculated as the average of the values of the rolling direction and the transverse direction of the coil. As for punching die wear, a newly produced punching die was used to punch the steel sheet and the wear was evaluated based on the change of the height of the burrs on the steel sheet in relation to the number of punching. Die with large wear results in large burrs of the steel sheet in a relatively small number of punching. The results are shown in Table 6 (continuation of Table 5).

As clear from the results shown in Table 6, the samples produced under the conditions of the present invention are soft and thus are good in rollability in the cold rolling process and result in small wear of the punching die before the precipitation heat treatment and become hard and superior in magnetic properties after

precipitation treatment.

Table 5

No.	Steel compositions (mass%)										Steel sheet heat history		Precipitation heat treatment			Metal phase composed of Cu before precipitation heat treatment	
	C	Si	Mn	P	S	Al	N	Cu	Other		Hot rolling *1	Hot rolling *2	Process	Temp-erature (°C)	Holding time (min)	Average size (μm)	Number density (/μm ³)
31	0.0025	1.11	0.48	0.044	0.0016	0.004	0.0008	0.007*	-	-	100	30	A	500	60	-	-
32	0.0061	1.02	0.50	0.076	0.0014	0.002	0.0016	0.02*	Ni:2.5 Nb:0.03	-	30	30	A	500	60	-	-
33	0.0007	1.11	0.49	0.052	0.0007	0.002	0.0006	0.53*	-	-	30	20	A	500	60	-	-
34	0.0016	1.03	0.46	0.048	0.0017	0.001	0.0025	0.89	-	-	60	60	A	500	60	0.11	0.1
35	0.0006	1.12	0.46	0.052	0.0005	0.004	0.0020	0.64	-	-	30	5	A	500	60	-	-
36	0.0021	1.01	0.50	0.061	0.0093	0.004	0.0028	0.88	-	-	40	5	A	500	60	0.004	0.01
37	0.0023	1.16	0.48	0.069	0.0310	0.004	0.0020	1.14	-	-	20	30	B	500	60	-	-
38	0.0074	1.16	0.48	0.069	0.0310	0.004	0.0020	1.14	-	-	20	30	B	500	60	-	-
39	0.0156	1.16	0.48	0.069	0.0310	0.004	0.0020	1.14	-	-	20	30	B	500	60	-	-
40	0.0013	1.08	0.53	0.043	0.0017	0.003	0.0017	2.40	Ca:0.005	-	60	30	A	500	60	-	-
41	0.0007	1.16	0.52	0.072	0.0011	0.004	0.0009	2.33	Ti:0.03	-	20	20	B	650	1	-	-
42	0.0006	1.17	0.52	0.071	0.0015	0.005	0.0021	2.83	-	-	30	20	A	400	300	0.03	0.15
43	0.0096	1.17	0.52	0.071	0.0015	0.005	0.0021	2.83	-	-	30	20	A	400	300	0.03	0.15
44	0.0020	1.11	0.49	0.066	0.0007	0.001	0.0025	4.25	Ni:1.9	-	20	10	A	350	300	0.06	0.2
45	0.0021	1.09	0.50	0.064	0.0013	0.001	0.0012	8.64*	-	-	30	30	A	350	300	1.3	0.03

Steel sheet heat history

*1: Residence time at temperature range of 450°C to 700°C in cooling process after final rolling of hot rolling (sec)

*2: Residence time in temperature range of 450°C to 700°C in cooling process in final annealing after cold rolling (sec)

Precipitation heat treatment

A: Cooling process in stress relief annealing process after working to motor

B: Only precipitation heat treatment after working to motor

* Constitution outside range of claims of present invention

Table 6

No.	After precipitation heat treatment of metal phase composed of Cu		Crystal grain average size (μm)	Mechanical properties before and after precipitation heat treatment and changes in same						Magnetic properties after precipitation heat treatment		Cold rollability	Wear of stamping die	Evaluation
	Average size (μm)	Number density ($/\mu\text{m}^3$)		Before	After	Comp.	Before	After	Diff.	B ₁₀ (T)	W _{10/400} (W/kg)			
31	-	-	100	121	118	0.97	391	361	-30	1.65	10.2	VG	G	C
32	-	-	70	200	154	0.77	630	465	-165	1.25	31.3	F	P	C
33	0.002	0.02		130	125	0.96	398	387	-11	1.60	15.5	VG	G	C
34	0.11*	0.1	120	118	120	1.02	378	363	-15	1.61	11.2	VG	G	C
35	0.004	10	150	133	166	1.25	422	502	80	1.69	11.8	VG	G	B
36	0.002	200	120	122	212	1.74	408	579	171	1.63	8.3	VG	G	A
37	0.004	200	100	146	241	1.65	443	616	173	1.65	9.2	VG	G	A
38	0.004	200	95	165	295	1.79	453	650	197	1.67	9.3	VG	G	A
39	0.004	200	100	177	312	1.76	460	670	210	1.68	9.1	VG	G	A
40	0.001	3000	70	123	187	1.51	386	538	152	1.62	9.2	VG	G	A
41	0.003	1000	70	149	211	1.42	457	600	143	1.59	11.4	VG	G	B
42	0.001	>10000	80	150	251	1.68	454	692	238	1.62	12.4	VG	G	A
43	0.001	>10000	85	165	292	1.77	460	725	265	1.65	12.0	VG	G	A
44	0.002	6000	70	135	277	2.05	440	784	344	1.56	12.0	VG	G	A
45	1.05*	0.11	70	135	146	1.08	420	420	0	1.43	24.5	G	G	C

Cold rollability

VG: Very good (no problem at all)

G: Good (fine adjustment required, but no problem)

F: Fair (sheet runnable if adjusting conditions)

P: Poor (large danger of sheet breakage)

Wear of stamping die

G: Good (small)

P: Poor (large)

Evaluation

A: Developed steel (very good)

B: Developed steel (good)

C: Comparative steel

(Example 4)

The steel compositions shown in Table 7 was made into a 250 mm thick slab and final sheets were produced by basically following process. The basic process conditions were a slab heating temperature of 1100°C, a final hot band thickness of 2.0 mm, a coiling temperature of 300°C or less in hot rolling, a hot band annealing of 980°C and 30 seconds, a final sheet thickness of 0.35 mm in cold rolling, and a recrystallization annealing temperature of the temperature or more at which the recrystallization occurred. After this, to simulate precipitation heat treatment after punching, a heat treatment at about 750°C was employed to control texture and precipitation of the metal phase. When the heat treatment simulates stress relief annealing, the precipitation heat treatment was employed in the cooling process after a heat treatment at 750°C for 2 hours. Each sheet before and after the heat treatments was measured for mechanical properties using a JIS No. 5 test piece and for magnetic flux density B_{10} and core loss $W_{10/400}$ using a 55 mm square SST test. The mechanical properties and magnetic properties were calculated as the average of the values of the rolling direction and the transverse direction of the coil. As for punching die wear, a newly produced punching die was used to punch the steel sheet and the wear was evaluated based on the change of the height of the burrs on the steel sheet in relation to the number of punching. Die with large wear results in large burrs of the steel sheet in a relatively small number of punching. The results are shown in Table 8 (continuation of Table 7).

As clear from the results shown in Table 8, the samples produced under the conditions of the present invention are soft and thus are good in rollability in the cold rolling process and result in small wear of the punching die before the precipitation heat treatment and become hard and superior in magnetic after the

precipitation treatment.

Table 7

No.	Steel compositions (mass%)										Steel sheet heat history		Precipitation heat treatment			Before precipitation heat treatment of metal phase composed of: Cu	
	C	Si	Mn	P	S	Al	N	Cu	Other		Hot rolling *1	Hot rolling *2	Process	Temp-erature (°C)	Holding time (min)	Average size (μm)	Number density (/μm ³)
46	0.0017	2.94	0.28	0.014	0.0010	0.47	0.0018	0.03*	-	-	40	20	B	550	10	-	-
47	0.0065	2.86	0.34	0.013	0.0015	0.49	0.0025	0.03*	Ni:2.5 Nb:0.03	-	30	10	B	550	10	-	-
48	0.0024	2.86	0.34	0.014	0.0007	0.54	0.0022	0.48*	-	-	100	20	B	550	10	-	-
49	0.0016	2.85	0.28	0.015	0.0014	0.54	0.0013	0.66	-	-	30	30	B	550	10	-	-
50	0.0013	2.88	0.34	0.019	0.0017	0.46	0.0011	0.96	-	-	30	20	B	550	10	0.05	0.02
51	0.0011	2.88	0.31	0.017	0.0004	0.50	0.0021	1.16	Se:0.012	-	120	20	B	550	10	0.04	0.01
52	0.0288	2.88	0.31	0.017	0.0004	0.50	0.0021	1.16	Se:0.012	-	120	20	B	550	10	0.04	0.01
53	0.0011	2.88	0.31	0.017	0.0004	0.50	0.0021	1.16	Se:0.012	-	120	20	A	550	20	0.04	0.03
54	0.0024	2.89	0.27	0.018	0.0016	0.53	0.0016	1.90	-	-	30	5	A	550	20	0.12	0.04
55	0.0023	2.91	0.25	0.011	0.0012	0.55	0.0028	1.93	Nb:0.03	-	20	10	B	600	5	-	-
56	0.0014	2.92	0.31	0.016	0.0007	0.45	0.0022	1.88	Ca:0.005	-	60	20	A	450	900	0.02	0.03
57	0.0005	2.87	0.29	0.014	0.0013	1.20	0.0022	2.60	-	-	900	10	A	450	300	0.35	0.2
58	0.0014	2.90	0.25	0.018	0.0005	1.05	0.0016	3.95	Ni:2.5	-	20	10	A	450	150	0.04	0.04
59	0.0010	2.91	0.31	0.013	0.0006	1.51	0.0028	4.89	-	-	20	5	B	720	0.3	0.06	0.02
60	0.0132	2.91	0.31	0.013	0.0006	1.51	0.0028	4.89	-	-	20	5	B	720	0.3	0.06	0.02
61	0.0023	2.91	0.32	0.019	0.0015	0.48	0.0029	8.33*	-	-	20	30	A	450	60	1.7	0.02

Precipitation heat treatment

A: Cooling process in stress relief annealing process after working to motor

B: Only precipitation heat treatment after working to motor

Steel sheet heat history

*1: Residence time at temperature range of 450 to 700°C in cooling process after final rolling of hot rolling (sec)

*2: Residence time in temperature range of 450 to 700°C in cooling process in final annealing after cold rolling (sec)

* Constitution outside range of claims of present invention

Table 8

No.	Precipitation heat treatment of metal phase composed of Cu		Crystal grain average size (μm)	Mechanical properties before and after precipitation heat treatment and changes in same					Magnetic properties after precipitation heat treatment		Cold roll-ability	Wear of stamping die	Evaluation
	Average size (μm)	Number density ($/\mu\text{m}^3$)		Before	After	Comp.	Before	After	Dif.	B_{50} (T)	$W_{15/500}$ (W/kg)		
46	-	-	150	176	182	1.04	530	530	0	1.65	2.5	VG	C
47	-	-	60	250	222	0.89	768	662	-106	1.20	8.7	F	C
48	0.01	0.1	130	168	166	0.99	530	552	22	1.68	2.7	VG	C
49	0.003	0.1	130	168	193	1.15	515	611	96	1.68	2.5	VG	B
50	0.002	20	170	174	220	1.27	543	710	167	1.71	2.2	VG	B
51	0.003	3000	130	171	255	1.49	529	783	254	1.69	2.3	VG	A
52	0.003	3000	110	213	322	1.51	583	830	247	1.70	2.5	VG	A
53	0.004	1000	120	171	267	1.57	529	779	250	1.68	2.5	VG	A
54	0.004	80	120	169	224	1.33	521	609	88	1.66	2.4	VG	B
55	0.001	>10000	90	167	280	1.68	518	816	298	1.62	2.7	VG	A
56	0.002	100	180	179	311	1.74	531	857	326	1.66	2.4	VG	A
57	0.006	100	100	218	256	1.17	690	744	54	1.64	3.0	F	B
58	0.003	4000	140	230	350	1.52	710	1150	440	1.65	2.2	G	B
59	0.002	>10000	150	171	358	2.09	515	1155	640	1.67	2.2	VG	A
60	0.002	>10000	150	201	408	2.03	604	1175	571	1.70	2.1	VG	A
61	0.76*	0.05	120	230	245	1.07	654	705	51	1.40	6.0	G	C

Cold rollability

VG: Very good (no problem at all)

G: Good (fine adjustment required, but no problem)

F: Fair (sheet runnable if adjusting conditions)

P: Poor (large danger of sheet breakage)

Wear of stamping die

G: Good (small)

P: Poor (large)

Evaluation

A: Developed steel (very good)

B: Developed steel (good)

C: Comparative steel

INDUSTRIAL APPLICABILITY

As explained above, the present invention enables stable production of a high strength electrical steel sheet which is hard and superior in magnetic properties. Furthermore, according to the present invention, it becomes possible to provide an electrical steel sheet having good workability at the time of processing to an electrical part and which is hard and good in magnetic properties at the time of use as an electrical part through a stable process conditions causing no refinement of texture and no trouble such as sheet breakage, by not allowing formation of almost any fine metal phase mainly composed of Cu in the steel sheet in the process of the production and by forming the metal phase in a heat treatment process after processing to an electrical part. Due to this, it becomes possible to secure strength, fatigue strength, and wear resistance without degrading the magnetic properties, and so greater efficiency, smaller size, superlonger lifetime, etc. of superhigh speed motors, motors incorporating magnets in rotors, and electromagnetic switch materials can be achieved.